# Investigation of the glass-crystal transition of $Na_5Ti(PO_4)_3$

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#### Abstract

 $Na_5Ti(PO_4)_3$  can exist in both crystalline and vitreous forms. The glass-crystal transition was investigated using several techniques. The crystallization process involves two steps corresponding to the successive formation of  $Na_4TiO(PO_4)_2$  and  $Na_5Ti(PO_4)_3$  crystals.

The latter new phosphate belongs to the Nasicon-type family  $(a_h = 9.061 \pm 0.002 \text{ Å}, c_h = 21.734 \pm 0.004 \text{ Å}, \text{ space}$ group R32). Its structure can be described as a three-dimensional framework containing TiO<sub>6</sub> and NaO<sub>6</sub> octahedra sharing common corners with the PO<sub>4</sub> tetrahedral groups. The titanium and sodium atoms are distributed with a 2-2 ordering along the c axis. This arrangement creates two different M<sub>2</sub> sites.

#### 1. Introduction

Transparent glasses with relatively large non-linearities are promising materials for all optical switching applications [1]. The choice of composition is mainly dictated by the possibility of an increase in the non-linear index coefficient  $n_2$ . Various parameters have to be considered such as high damage threshold, ultrafast response time, wave guide formation process *etc*.

A preliminary investigation of the Na<sub>2</sub>O-TiO<sub>2</sub>-P<sub>2</sub>O<sub>5</sub> system was recently reported [2]. P<sub>2</sub>O<sub>5</sub> was selected as a glass-forming oxide, and TiO<sub>2</sub> as the hyperpolarizable entity, and sodium ions were introduced to make the possible ion exchange easier. The glass-forming region was found to comprise the composition 50% Na<sub>2</sub>O, 20% TiO<sub>2</sub> and 30% P<sub>2</sub>O<sub>5</sub> (mole per cent). The crystallization of this glass leads to the formation of a new Nasicon-type phosphate of identical composition. The present paper describes the process of crystallization as well as the structure of the crystalline phase.

#### 2. Experimental details

Glasses were prepared as described elsewhere [2]. The crystallized form of  $Na_5 Ti(PO_4)_3$  is obtained by heating the glass of identical composition at 700 °C or by standard solid state preparation from a mixture of  $Na_2CO_3$ ,  $TiO_2$  and  $(NH_4)_2HPO_4$ .

The reaction

 $5Na_2CO_3 + 2TiO_2 + 6(NH_4)_2HPO_4 \longrightarrow$ 

 $2Na_5Ti(PO_4)_3 + 12NH_3 + 9H_2O + 5CO_2$ 

is completely achieved after three thermal treatments respectively at 300 °C for 4 h, 500 °C for 6 h and 750 °C for 15 h.

Slow cooling of the liquid of identical composition from 850 °C to room temperature (RT) leads to the simultaneous formation of  $Na_4 TiO(PO_4)_2$  and  $Na_5 Ti(PO_4)_3$  crystals. Suitable crystals of the latter phase were isolated for a complete structural determination.

#### 3. Results

The differential thermal analysis curve shows two glass transition temperatures  $T_{g1}$  and  $T_{g2}$  respectively at 414 and 507 °C, and two exothermic peaks  $T_{c1}$  and  $T_{c2}$ at 462 and 547 °C. The melting point is observed at about 800 °C (Fig. 1). The X-ray diffraction (XRD) patterns of samples heated between  $T_{c1}$  and  $T_{c2}$  reveal the existence of Na<sub>4</sub>TiO(PO<sub>4</sub>)<sub>2</sub> and Na<sub>4</sub>P<sub>2</sub>O<sub>7</sub> in addition to a remaining vitreous phase. Above  $T_{c2}$  only the new phase Na<sub>5</sub>Ti(PO<sub>4</sub>)<sub>3</sub> is detected. Therefore,  $T_{c2}$  corresponds to crystallization of the last amorphous phase and to the formation of the new phosphate.



Fig. 1. Differential thermal analysis curve of the 50 mol% Na<sub>2</sub>O, 20 mol% TiO<sub>2</sub>, 30 mol% P<sub>2</sub>O<sub>5</sub> glass.

#### 4. The structure of $Na_5Ti(PO_4)_3$

Here we summarize the description of the structure of  $Na_5Ti(PO_4)_3$ , details of which will be published elsewhere [3].

At room temperature the XRD pattern of Na<sub>5</sub>Ti(PO<sub>4</sub>)<sub>3</sub> can be indexed assuming a hexagonal cell  $(a_h = 9.061 \pm 0.002 \text{ Å}, c_h = 21.734 \pm 0.004 \text{ Å}, Z = 6, d_{exp} = 2.84, d_x = 2.89).$ 

The space group R32 is compatible with the observed reflections (hkl), -h + k + l = 3n and (hhl): l = 2n, as well as with a positive test of second harmonic generation (SHG = 1.064  $\mu$ m/0.532  $\mu$ m).

The structure was solved using standard diffraction techniques and calculations. It consists of a threedimensional network of PO<sub>4</sub> tetrahedra and AO<sub>6</sub> octahedra (A  $\equiv$  Ti, Na) sharing common corners. These octahedra are occupied successively by the titanium and sodium atoms with a 2-2 ordering along the *c* axis (Fig. 2).

The site usually labelled  $M_1$ , an elongated antiprism, shares common faces with a TiO<sub>6</sub> octahedron and an NaO<sub>6</sub> octahedron, giving rise to infinite chains parallel to the [001] direction with the following distribution:

# $[TiO_6][TiO_6][NaO_6(M_1)][NaO_6][NaO_6]$

 $[NaO_6(M_1)][TiO_6]$ 

This cationic ordering creates two different  $M_2$  sites within the three-dimensional framework where the remaining sodium atoms are located: they are surrounded by eight oxygen atoms with relatively large Na-O distances in the range 2.437-2.904 Å.

These data can be analysed in the context of the structural evolution of the Nasicon-type phosphates.



Fig. 2. A perspective drawing of the Na<sub>5</sub>Ti(PO<sub>4</sub>)<sub>3</sub> structure.

This structure was described initially by Hagman and Kierkegaard [4] for  $NaA_2(PO_4)_3$  (A = Ti, Ge, Zr) phosphates in which only the M<sub>1</sub> site is occupied by the sodium atom.

The rules governing the variations in cell parameters of the Nasicon-type phosphates have already been established [5]: (i) the *a* parameter is directly correlated with the size of the cation located at the octahedral A site and with the proportion of sodium in position  $M_2$ ; (ii) the *c* parameter increases with the size of the  $A^{n+}$  ion and the strength of the Na<sup>+</sup>(M<sub>1</sub>)-A<sup>n+</sup> electrostatic repulsion.



Fig. 3. Cell parameters and volume variations vs. mean A ionic radii of sodium-rich Nasicon-type phosphates.

Figure 3 compares the variation in structural data of Nasicon-type phosphates for which the mean charge of the A cations is 2.5. The linear variation of these parameters vs. the mean A cation ionic radii is in fair agreement with the cationic distribution found for  $Na_5Ti(PO_4)_3$ .

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